## **TEM Analysis of Multivalent Ion Battery Cathode**

Bilash KC<sup>1</sup>, Jinglong Guo<sup>1</sup>, Robert Klie<sup>1</sup>, D. Bruce Buchholz<sup>2</sup>, Guennadi Evmenenko<sup>2</sup>, Jae Jin Kim<sup>3</sup>, Timothy Fister<sup>3</sup> and Brian Ingram<sup>3</sup>

<sup>1</sup>University of Illinois at Chicago, Chicago, Illinois, United States, <sup>2</sup>Northwestern University, Evanston, Illinois, United States, <sup>3</sup>Argonne National Lab, Lemont, Illinois, United States

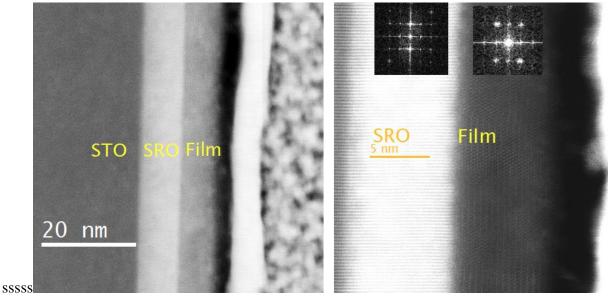
The rechargeable batteries as of today are predominantly Li-ion batteries (LIBs) [1]. LIBs capacity, however, is intrinsically limited by the monovalency of Li<sup>+</sup> ions. The increasing demand for rechargeable batteries with high energy and power density has led to exploration of several alternatives to LIBs, such as multivalent Mg ion batteries. Since Mg<sup>2+</sup> ions are comparable in size to Li<sup>+</sup> ions and can store twice the charge upon intercalation than Li<sup>+</sup>, Mg ion battery is an attractive option for high energy density [2]. However, due to difficulty of Mg intercalation into the host, need for suitable combination of electrolyte and cathode structure need to be identified [3]. Drawing upon the success of LiMn<sub>2</sub>O<sub>4</sub>, the spinel oxide Mn<sub>2</sub>O<sub>4</sub> host is examined as suitable intercalation structure for Mg<sup>2+</sup> with higher theoretical capacity (270 mAh/g for MgMn<sub>2</sub>O<sub>4</sub>) [4]. Although many studies have shown signs of reversible Mg intercalation, the amount of Mg intercalation have been very low and non-uniform and is plagued by limited understanding of various phenomena such as intercalation sites, Mg-Mn inversion, high migration barriers [5]. Moreover, most of these studies are based on polycrystals or powder samples which are difficult to interpret. By isolating a specific orientation, single-crystal cathodes allow for detailed characterization of surfaces and interfaces, defects concentrations, depth profiling of chemical and structural changes to examine the (de)intercalation process of Mg ions.

In this contribution, we will describe our atomic-resolution characterization study of Mg intercalation in single-crystal thin-film Mn<sub>2</sub>O<sub>4</sub> host structures. Single-crystal LiMn<sub>2</sub>O<sub>4</sub> thin films were grown ons SrTiO<sub>3</sub> with SrRuO<sub>3</sub> as conductive buffer layer using pulsed laser deposition. The films and buffer layer follow epitaxial relationship of LiMn<sub>2</sub>O<sub>4</sub> (111) // SrRuO<sub>3</sub> (111) // SrTiO<sub>3</sub> (111) out of plane. Li ions were deintercalated by charging to 4.3 V leaving behind Mn<sub>2</sub>O<sub>4</sub> framework. Mn<sub>2</sub>O<sub>4</sub> was then used as host cathode structure with Mg anode and 0.05 M Mg(TPFA)2 (Magnesium Perfluoroalkoxyaluminate) in triglyme electrolyte build into custom designed X-ray transmission cell for electrochemistry of Mg ions intercalation, and discharged from 4.3V to 2.0 V. SrRuO<sub>3</sub> maintained conductivity and crystallinity within the cycling voltage range of 2.0-4.3 V

Figure 1a) and b) show images of ~ 10 nm of delithiated LMO and subsequently Mg intercalated Mg-Mn<sub>2</sub>O<sub>4</sub> film taken along the [110] zone axis using JEM-ARM200CF scanning transmission electron microscope (STEM) at UIC. JEM-ARM200CF is probe aberration corrected with a cold field emission source with 0.35eV energy resolution, and spatial resolution of less than 0.08nm for high-angle annular dark field (HAADF) imaging at an acceleration voltage of 200 kV. Oxford Energy Dispersive X-ray Spectroscopy (XEDS) signal was recorded using Oxford X-Max 100TLE SDD X-ray detector. We find that the Mg- Mn<sub>2</sub>O<sub>4</sub> film is continuous and maintains good epitaxy close to the interface with the SrRuO<sub>3</sub> buffer layer. However, near the thin film surface, we find an amorphous layer 2-3 nm thick. Figure 2 shows the EDS mapping of the magnesiated film. We find that Mg shows the highest concentration in the



amorphous surface layer and very little Mg was found inside the crystalline film. This difficulty for Mg intercalation in polycrystal and nano powders has been well reported in literature [3,4,5] and initial analysis of single crystal thin films clearly demonstrates the challenge. Detailed examination of the electrochemistry and optimal conditions for Mg intercalation is under investigation.[6]



**Figure 1.** (a). LAADF image shows SrRuO3 buffer layer and Mg- Mn2O4 layer appear continuous. (b) HAADF image shows SrRuO3 is epitaxial and Mg- Mn2O4 appears polycrystalline. Insets show epitaxial and crystalline nature of SRO and film.

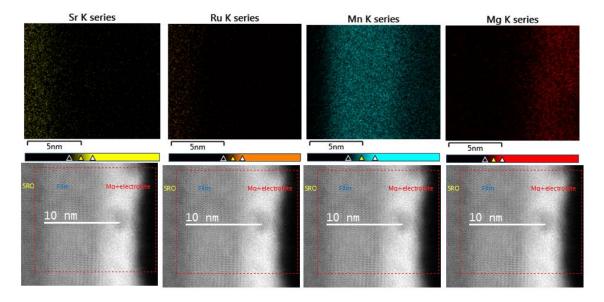


Figure 2. Top layer of the Mg-Mn2O4 appear amorphous. Mg is detected in top amorphous layer

## References

[1] Manthiram, A., Yu, X., and Wang, S., Nature Reviews Materials 2.4 (2017), p. 16103.

- [2] Yoo, H. D., Shterenberg, I., Gofer, and Y., Gershinsky. Energy & Environmental Science 6 (2013), p. 2265-2279.
- [3] E. Levi, Y. Gofer, and D. Aurbach, Chemistry of Materials 22 (2010), p. 860
- [4] Ichitsubo, T., Adachi, T., and Yagi, S., J. Mater. Chem. 21 (2011), p.11764
- [5] Malavasi, L., Ghigna, P., and Chiodelli, J. Solid State Chem. 166.1 (2002), p.171
- [6] This work was support by the Joint Center for Energy Storage Research (JCESR), an Energy Innovation Hub funded by the U.S. Department of Energy, Office of Science, Basic Energy Sci