## X-ray Microanalysis of Insulators in a Variable Pressure Environment

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In a low vacuum environment, electric fields generated by ionized gas molecules and sub-surface trapped charge (Q) can alter the primary electron landing energy ( $\epsilon^{DHL}$ ). Consequent artifacts in x-ray microanalysis can be alleviated by working under conditions whereby the net electric field (<u>E</u>) is dominated by the component produced by gaseous ions (<u>E</u><sup>ION</sup>), and excess ions are rapidly removed *via* efficient ion neutralization routes. Such conditions can be attained over a wide of microscope operating parameters simply by employment of appropriate sample-electrode geometries.

In a variable pressure/environmental SEM, the electric field at each point (x,y,z) in the space between the sample and the pole piece typically consists of three distinct components:

 $\underline{E}(x,y,z) = \underline{E}^{E}(x,y,z) + \underline{E}^{Q}(x,y,z) + \underline{E}^{ION}(x,y,z)$ (1) where  $\underline{E}^{E}$  is the field generated by a biased electrode (ie, the electron collector of a gaseous electron detector [1]) and  $E^{Q}$  is the field produced by Q. If the specimen is an uncoated insulator,  $\varepsilon^{DHL}$  and the maximum bremsstrahlung x-ray energy (the Duane-Hunt limit, DHL) are given by:

 $\varepsilon^{DHL} = eV^{A} + e(\Delta V^{E} + \Delta V^{Q} + \Delta V^{ION}) = eV^{A} + e\Delta V$  (2) where e is the charge of an electron,  $V^{A}$  is the primary electron accelerating voltage, and  $\Delta V$  is the net potential difference between the pole piece and the sample surface corresponding to  $\underline{E}^{E}$ ,  $\underline{E}^{Q}$  and  $\underline{E}^{ION}$ . The sign of  $\Delta V^{Q}$  is determined by the net polarity of sub-surface charge, as in the case of high vacuum SEM [2,3]. However, in contrast to high vacuum SEM,  $\Delta V^{ION}$  can cause a significant increase in  $\varepsilon^{DHL}$ , alter the overvoltage and compromise x-ray quantification procedures. Such increases in  $\varepsilon^{DHL}$  are illustrated by the energy-dispersive x-ray spectra shown in Fig. 1, acquired as a function of electrode bias ( $V^{E}$ ) and gas pressure (P). The data clearly illustrate that  $\varepsilon^{DHL}$  scales with  $V^{E}$  and P. This behavior is attributed to the influence of  $V^{E}$  and P on  $\underline{E}^{E}$ ,  $\underline{E}^{ION}$  and  $\underline{E}^{Q}$ , and consequent effects of  $\Delta V$  on  $\varepsilon^{DHL}$  (see Eqn. 2). We will present a detailed model of  $\varepsilon^{DHL}$  behavior in a low vacuum environment, based on knowledge of the polarity of  $\Delta V^{Q}$  and  $\Delta V^{ION}$ , obtained from simultaneous measurements of x-ray spectra and Q-induced changes in the SE emission current.

From a practical viewpoint, it is desirable to eliminate the changes in  $\varepsilon^{DHL}$  caused by  $\Delta V$  without imposition of restrictions on operating parameters such as  $V^A$ ,  $V^E$ , working distance and gas pressure. On the basis of the aforementioned model, this can be achieved if: (i)  $\Delta V^E$  is minimized, (ii)  $\Delta V^{ION} > \Delta V^Q$ , and (iii) excess ions are rapidly neutralized so that:  $\Delta V^{ION} + \Delta V^Q \approx 0$ . In a low vacuum environment such conditions can be attained simply by employment of appropriate sample-electrode geometries in the specimen chamber. Equipotentials calculated for two-dimensional representations of two such geometries are shown in Fig. 2. The effectiveness of these geometries in reducing  $\Delta V$ -induced  $\varepsilon^{DHL}$  shifts is demonstrated by the corresponding x-ray spectra also shown in Fig. 2. These results will be explained using the abovementioned model. [4]

References

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FIG. 1. Energy-dispersive x-ray spectra acquired consecutively from the same region of mica, in the order shown in the figures, as a function of electrode bias ( $V^{E}$ ) and water vapor pressure (P): (a) P = 10<sup>-6</sup> torr, (b) P = 0.2 torr. The data show that  $\varepsilon^{DHL}$  scales with  $V^{E}$  and P [ $V^{A}$  = 1 kV, working distance = 10 mm, electrode-sample separation = 4.5 mm, horizontal field width = 130 microns].



FIG. 2. X-ray spectra of mica showing differences between  $\varepsilon^{DHL}$  in data acquired when a ring electrode was placed 4.5 mm above the sample ("standard" geometry) and: (a) an array of grounded Cu wires was placed 0.55 mm above the specimen, or (b) the ring electrode was replaced with an off-axis plate electrode. [V<sup>A</sup> = 2 kV, P = 0.5 torr, working distance = 10 mm, horizontal field width = 130 microns]. The insets show the electric equipotentials (broken lines) calculated for simplified two-dimensional representations of the abovementioned geometries [V<sup>E</sup> = 500 V]. The spectra show that employment of these geometries serves to reduce Duane-Hunt shifts caused by gaseous ions.