Transmission EELS Quantification Study of Oxygen Content at Zirconium Alloy Metal/Oxide Interface

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A great deal of careful experimentation has been carried out over the past 40 years on the critical mechanisms of oxidation in the zirconium alloys used for nuclear fuel cladding. However, there is still no consensus on how the nano-scale structure and chemistry of the metal/scale interface influence the oxidation kinetics, although this information is vital for the accurate prediction of the performance of existing alloys and the design of new materials for higher burn-up regimes. The presence of an oxygen-rich region at the metal/oxide interface has been previously reported as a suboxide phase [1-3]. Since oxidation of Zr alloys proceeds primarily by inward diffusion of oxidizing species, such a sub-oxide could play a crucial role in controlling oxidation kinetics, but there is still considerable disagreement on the precise composition and structure of this layer. Previous quantitative analysis of light elements such as oxygen. In this work we show that reliable quantification results on the oxygen content at the metal/oxide interface can be obtained by electron energy-loss spectroscopy (EELS) analysis when employing the most up-to-date TEM sample preparation and analysis techniques.

TEM thin foils were prepared from oxides on Zirconium alloys by employing the *in situ* lift-out FIB method [4, 5] on a FEI FIB 200 instrument (5-30kV Ga⁺ incident beam energy with currents of 50-5000 pA). This technique has the advantage of producing large, homogenous, electron transparent foils less than 100 nm thick containing the metal/oxide interface as required for EELS. TEM and EELS analysis were carried out on a Jeol 3000F microscope, equipped with a Gatan GIF and operated at 297 kV.

A series of calibrations were necessary to achieve reliable quantitative EELS data. First, a new experimental ratio between Zr and O inelastic scattering cross-sections ($\sigma_{zr}/\sigma_o \approx 7.27$) was obtained using a stoichiometric ZrO₂ powder sample. After this calibration, the measured oxygen content in the bulk oxide was found to be very close to the 67 at. % theoretical oxygen content of ZrO₂, compared to ~80 at. % before calibration. Secondly, to evaluate the influence of multiple scattering, the inelastic mean free path, λ , under our experimental conditions (α =11.5mrad and β =9mrad) was estimated by acquiring a series of EELS spectra from a specially prepared wedge-shaped TEM specimen (1° angle) (Figure 1). The absolute thickness was determined from a HAADF line profile along the line shown in Figure 1b . A linear fitting of t/λ vs. t yielded an experimental value of ~99nm for λ , significantly smaller than the calculated value of 115 nm [6]. Quantitative results from the wedge-shaped sample confirmed that the expected Zr/O ratio for the ZrO₂ far from the interface was obtained for sample thicknesses below 80nm.

A commercial Zircaloy-4 recrystallized sheet (1.24% Sn, 0.17% Fe, 0.1% Cr, w/w) tested in an autoclave at 360 °C for 54 days developed a 1.2-1.5 μ m pre-transition oxide and was characterized using our new methodology. The HAADF image in Figure 2a shows a distinct layer of intermediate contrast ~50 nm thick between the metal and the columnar oxide. The EELS line profile (figure 2b)

across the same interface reveals an intermediate suboxide layer of similar thickness with a relatively constant oxygen content of ~ 45 % (as opposed to the ~66% oxygen measured in the bulk oxide). We are using this methodology to explore whether these suboxide layers may explain some of the observed differences in corrosion performance between different Zr-alloys.

References

- [1] A. Yilmazbayhan et al., Journal of Nuclear Materials 349 (2006) 265.
- [2] P. Bossis et al, Zirconium in the Nuclear Industry: 12th International Symposium, 2000, 918.
- [3] S. Abolhassania et al., Journal of Nuclear Materials Article in Press (2009).
- [4] J. Mayer et al., MRs Bulletin 32 (2007) 400.
- [5] S. Lozano-Perez, Micron 39 (2008) 320.
- [6] T Malis et al., Journal of Electron Microscopy Technique 8 (1988) 193.

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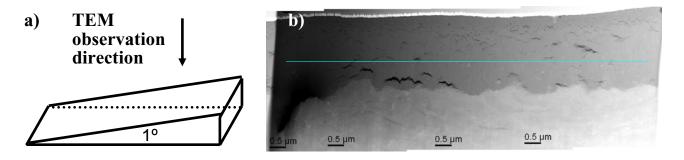


Figure 1: a) Schematics of the FIB prepared wedge sample. b) HAADF image of the wedge sample.

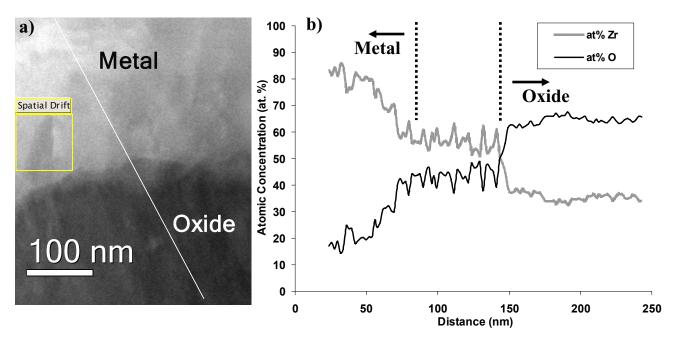


Figure 2 a) HAADF image of the metal/oxide interface region and b) oxygen concentration line profile across the metal/oxide interface showing the presence of a sub-stoichiometric oxide layer.