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Abstract: International standard ISO-25498 specifies the method of selected area electron diffraction (SAED) analysis in TEM. It is applicable to the acquisition of SAED patterns, indexing the patterns and calibration of diffraction constant. Several features of this standard are introduced. As an example of the applications, phosphide precipitates in the steel are identified as $M_xP$, where $x$ is 2–3 and $M$ is Fe, Ti, Cr, or Ni. It possesses a hexagonal lattice with lattice parameter $a = 0.609 \text{ nm}$ and $c = 0.351 \text{ nm}$.

Key words: international standard, electron diffraction, TEM, phosphide precipitate, CSP steels, EDS microanalysis, nanoparticles

INTRODUCTION

The international standard ISO-25498 (published in 2010) specifies the method of selected area electron diffraction (SAED) analysis in TEM. This standard was prepared by Technical Committee ISO/TC 202 (microbeam analysis)/SC3 (analytical electron microscopy). Experts from nine countries of the member bodies have participated in the work.

The standard ISO-25498 is applicable to the acquisition of SAED patterns from crystalline specimens, indexing the patterns and calibration of diffraction constant. It is also a supplementary technique for the acquisition of high-resolution images, microdiffraction, or convergent beam diffraction studies. The present contribution introduces some features of ISO-25498 briefly. An example of the applications is also given.

MATERIALS AND METHODS

As an example of the applications, phosphide $M_xP$ particles were analyzed by SAED and EDS. Composition of the experimental steel is: (mass%) C: 0.06, Si: $\leq 0.6$, Mn: $\leq 0.6$, S: $\leq 0.01$, P: $\leq 0.12$, Cu: $\leq 0.3$, Cr: $\leq 0.5$, Ni: $\leq 0.2$, Ti: $\leq 0.13$. The steels were produced by compact strip production (CSP) processing. Specimens were extractive powder, which were extracted electrochemically from the steel then spread on the microgrids.

Features of the ISO-25498

Optimizing the Optical Coupling of TEM for SAED Analysis

In a TEM, diffraction pattern appears in the back focal plane of the objective lens (OL). This pattern is magnified by the intermediate and projector lenses, and displayed on a viewing screen. The interest area is selected by the field limiting aperture (FLA), which is inserted at the imaging plane of the OL. Therefore, this specimen area is actually selected from its image. In SAED working mode, the object plane of the intermediate lens (IL) coincides with back focal plane of the OL, but in the imaging mode this object plane of the IL coincides with the imaging plane of the OL (see Fig. 1). Discrepancy of the selected area may arise from spherical aberration (coefficient $C_s$) of the OL and the specimen height. The selected area in the specimen will be shifted by distance $\gamma$:

$$\gamma = C_s(2\theta_b)^3 + 2\theta_b \cdot \Delta f$$

where $\Delta f$ is the deviation of the specimen height from the focusing position and $\theta_b$ is Bragg angle of the reflection beam.

Thus, the diffraction information actually comes from a specimen area, which is obviously larger than the area constrained by the FLA aperture. Objects out of the selected area may also have contribution to the SAED pattern. It can be seen that the specimen height is critical to ensure that a diffraction pattern is corresponding to the area selected by the FLA aperture. The minimum analyzable area for the SAED method is limited by OL aberration. The attempt of the operating procedure described in ISO-25498 is to optimize optical coupling of these lenses, FLA aperture, and the specimen. Then credible results could be received.

Obtain Information from Three-Dimensional Space of the Specimen

Information from three-dimensional space of the specimen is essential for diffraction study. As a single diffraction pattern consisting of only zero Laue zone will provide only planar (two-dimensional) information. It is necessary to acquire additional diffraction patterns from the same area but with different crystal orientation or from different grains/
particles of the same phase. Two methods, the specimen tilting and the multicolony procedure, are specified by the standard.

1. The specimen tilting procedure: choose a row of close-spaced diffraction spots collinear with the central spot on the diffraction pattern. Align this row with the tilting axis and obtain the second (and more) diffraction pattern by tilting the specimen.

2. The multicolony procedure: acquire the second or more patterns from several areas, i.e., different particles or grains of the same phase. In this situation, it is necessary to choose appropriate diffraction spots to form a dark-field image to confirm the source of each diffraction pattern. Further confirmation may be achieved through simultaneous analysis of EDX.

The three-dimensional reciprocal space of a specimen may be constructed by these patterns.

**Indexing Diffraction Pattern by the Characteristic Parallelogram**

The success of the SAED method relies on the validity of indexing the diffraction patterns. Concept of the characteristic parallelogram is introduced into ISO-25498 for indexing of the patterns (Edington, 1975; Guo et al., 1983). A diffraction pattern consisting of only zero Laue zone from a single crystal appears as an array of “spots,” the basic unit of which is characterized by a basic parallelogram (Fig. 2). This parallelogram is constituted by the vectors $R_1$ and $R_2$. These vectors are corresponding to the nearest and next nearest diffraction spots $h_1k_1l_1$ and $h_2k_2l_2$ to the central spot, respectively, in the pattern. Work out the interplanar spacing corresponding to the diffraction spots $h_1k_1l_1$ and $h_2k_2l_2$ in the pattern. Then the diffraction pattern can be indexed by comparing with the characteristic parallelogram and the zone axis can be decided. The method of indexing the diffraction pattern by using the characteristic parallelogram is recommended by ISO-25498.

**Improvement in Accuracy of the SAED Studies**

After a SAED pattern has been obtained and indexed correctly, a successful analysis also depends on the fact that the interplanar spacing of each reflection in the pattern can be measured properly. This measurement involves the use of a reference material so that the camera constant can be determined.

Sources of error in determining interplanar spacing include fluctuations in the voltage supply and lens current of the imaging lenses, shift in specimen height, and the aberration from the lenses. The effects of these parameters are discussed in ISO-25498. Provided the experimental conditions for acquiring the diffraction pattern of the standard and the unknown specimen are completely identical, the effects of variation in microscope parameters are very small. Thus, the main errors will come from measurement of the diffraction pattern. Effects of the lens aberration, especially the astigmatism, are crucial. In ISO-25498, the procedure for determination of the camera constant and the interplanar spacing is specified to minimize the errors.

**Identification of Phosphide M$_x$P by SAED and EDX Analysis**

As the solidification rate and cooling rate of the steels in the CSP processing is highly rapid, supersaturation of the elements such as Ti, O, S, N, P, etc. is much increased in the steels. Precipitation behavior of the second phases can be varied obviously in the steels (Liu, 2009). A large number of precipitates with nanometers in size such as sulfides, oxides, and carbonitrides were observed. However, phosphide particles with nanometer scale in low-carbon steels are seldom reported.

Recently, phosphide precipitates in the experimental steel have been investigated by SAED and EDX (Lou & Liu, 2010). The results showed that phosphide particles M$_x$P ($x = 2–3$) in nanometer scale exist in the steels. The value of $x$ is 2–3; metallic elements M are Fe, Ti, Cr, or Ni. Figure 3 shows a TEM image of a rod-shaped phosphate particle (Fe, Ti)$_x$P. The EDX and SAED patterns of the particle are given.
**SUMMARY**

The ISO-25498 (2010) is the first standard for SAED analysis in TEM. Only fundamental procedures are specified in this standard. Further improvement is in prospect. All of the applications are based on the fact that the SAED patterns are correctly acquired, indexed, and measured. Correct operation for the electron diffraction study relies on comprehensive understanding of the microscopy principle. Phosphide M$_x$P ($x=2$–3) precipitates in nanometer scale in the CSP steel containing titanium and phosphor are identified by SAED and EDX. The results approved that the phosphide M$_x$P can present as dispersive precipitates in the steels produced by CSP process.

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**REFERENCES**


