Atomic-scale Observation of Grains and Grain Boundaries in Monolayers of WS$_2$

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Transition metal dichalcogenides (TMDs), i.e. MoS$_2$ and WS$_2$, are fascinating materials due to their optical, electronic and catalytic properties. Monolayers of semiconducting TMDs possess a direct band gap and large optical absorption [1,2] ideal for constructing novel optical and electronic devices. In addition, MoS$_2$ and WS$_2$ have a high catalytic activity, and a wide range of physiochemical properties [3,4]. Such applications arise from the scalable synthesis of atomically thin TMD sheets via chemical vapor deposition (CVD). Grain boundaries (GB) and dislocations are inevitable defects in 2D layered crystals synthesized by the CVD [5]. Therefore, it is important to image and probe defects and their atomic and chemical structure before they can be nano-engineered.

Because of the unique three-atomic thickness in monolayers of TMDs, defects behave very differently from those observed in 2D monatomic films (i.e. graphene). In addition, this sandwich structure can lead to unique dislocations and different grain boundary structures and dynamics. We use ultra-high resolution aberration-corrected electron microscope (TEAMI) with sub-Ångstrom resolution in order to understand the atomic structure of GB and dislocations in WS$_2$ monolayers. Annular dark field (ADF) STEM images of GB in WS$_2$ show the atomic structure of the CVD-grown films [6]. The material contains monolayer, bilayer and trilayer regions of WS$_2$ with different stacking orders, and the layers display a slight rotational mismatch with respect to each other (Fig 1A). At 80 kV, we observe W atoms to be highly stable under the electron beam, while S atoms show significant movement around the edges and near the dislocation cores (see Fig 1). We investigated the structure of GB in monolayered WS$_2$ as well as their stability and dynamics under the electron beam. Figure 1D shows a tilt GB in monolayered WS$_2$ with a few edge dislocations along the GB. We identify that GB have a 6|8 dislocation component, which is consistent with that reported in predictions on TMDs under S-rich conditions, rather than the formation of the 5|7 dislocation [7]. This study further employs first principles calculations to elucidate the dislocation migration mechanisms in monolayers of WS$_2$. Our experimental observations coupled with theoretical simulations indicate that it is possible to achieve a high dislocation mobility in WS$_2$ which leads to the efficient reconstruction of GB.
References:

Figure 1. (A) The annular dark-field (ADF)-STEM image of a WS2 film with monolayer, bilayer and trilayer regions, (B) Atomic-resolution ADF-STEM image of a single-layer WS$_2$, indicating W and S atoms, (C) the top and side view of a monolayer WS2 film, and (D) ADF-STEM image of a tilt grain boundary in a single-layer WS$_2$. 

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